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PROPERTIES OF GOLD NANOANTENNAS IN THE INFRARED

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ith infrared spectroscopic microscopy using synchrotron light, resonant light scattering from single gold nanowires was investigated. The nanowires with diameters in the range of 100 nm were prepared by electrochemical deposition in polycarbonate etched ion-track membranes and transferred onto infrared-transparent substrates. For a few microns long nanowires, antenna-like plasmon resonances were observed in agreement with light scattering calculations. The resonances are dependent on wire length, wire shape, and on the dielectric surroundings of the nanowire. Electromagnetic far-field enhancement at resonance points to the ability of the nanowire to confine light on the nanoscale. This effect can be exploited for surface enhanced infrared absorption.

12.1 Introduction

Below the onset of interband transitions, the infrared (IR) optical properties of metals are determined by the collective oscillations of free

charge carriers called plasmons. The Drude dielectric function

$$\varepsilon(\omega) = \varepsilon_{\infty} - \frac{\omega_{\rho}^{2}}{\omega(\omega + i\omega_{\tau})}$$
(1)

is a good description of that behaviour.¹⁻⁵ Effects from interband transitions on the background polarizability are included in ε_{∞} . The Drude parameters ω_{p} and ω_{τ} describe the plasma frequency and the relaxation rate of the free charge carriers, respectively. With ω_{p} much higher than IR frequencies and $\omega_{\tau} \ll \omega_{p}$ the Drude dielectric function has a strong negative real part in the IR and corresponds to a considerably high polarizability. Above (circular) frequency $\omega = \omega_{p} / \sqrt{\varepsilon_{\infty}}$ (the so-called plasma edge) the high metallic reflectivity drops down to values that depend on background polarizability and interband transitions. The high reflectivity at lower frequencies is related to the small penetration depth (skin depth) of the IR light into the metal. For ω_{τ} in the mid IR (typical for iron or defect-rich noble metals) the skin depth in the IR⁶ is

$$\delta = \frac{c}{\omega_{\rm p}} \sqrt{\frac{2\omega_{\tau}}{\omega}} \tag{2}$$

with c as the velocity of light. Typical values for skin depths in the mid IR are 10 to 50 nm for metals. Therefore, metal films of a few nm in thickness are partially transparent in the IR. This behaviour extends to the visible range where (for sufficiently large ω_{n}) in case of oblique incidence of p-polarized light the transmittance spectra of ultrathin metal films show the additional plasmon resonance around $\omega = \omega_{p} / \sqrt{\varepsilon_{\infty}}$ due to the dispersion of surface plasmon polaritons.⁷ This resonance corresponds to collective charge carrier oscillations (plasmons) in perpendicular direction for which the film boundaries are effective. For metal particles the size and shape determine the boundary conditions for the plasmon oscillations. Vice versa, the optical resonances can be tuned via shape and size. The tunable optical properties of metal nanoparticles and their applications have been a topic of dramatically increasing interest over the last years⁸⁻¹¹. In particular, this interest is related to the considerable electromagnetic field enhancement in the near-field of a metal nanoparticle at resonance. As an example, it can be used to control field amplification in surface enhanced Raman scattering (SERS)^{12,13}. Extremely high near-field enhancement was calculated for the region of a

slit between nanorods¹⁴. Related to this, electromagnetic resonances of nano-objects can be exploited to confine light to a volume of subwavelength dimensions^{15,16}. Arrays of nanosized objects exhibit interesting photonic properties¹⁷ and composites of nanostructures are the focus of search for metamaterials¹⁸⁻²⁰. Compared to antenna-like resonances in the visible range²¹, electromagnetic field enhancement by metal antennas is higher in the infrared because of the strong negative part of the metal dielectric function and the high aspect ratios that can be produced. The resonant modes of collective charge carrier motion that can be excited by electromagnetic radiation with wavelength much larger than the particle size are mainly determined by the material properties of the nanoparticles.²² Going from spherical particles to prolate spheroids, the shape dependence of the resonances becomes much $stronger^{23,24}$ mainly for electric field along the longer particles axis. The plasmon resonance splits into branches with a low-energy mode, which can be detected in the mid-infrared for an adequate length of the nanoparticle²⁵. The larger the aspect ratio (long axis L / small axis D) of the particle the lower the frequency of this mode. When the wire length approaches the range of the exciting wavelength, the effects of retardation dominate the resonance condition and the limited speed of light leads to a direct dependence of the resonance frequency on the absolute size of the particle. This is well known from purely classical theory of scattering of electromagnetic waves by ideal metal objects²⁶. Here antenna resonances occur if the length L of an infinitely thin wire roughly matches with multiples of the wavelength λ , *i.e.*

$$L = \frac{\lambda}{2n} \cdot m \,, \tag{3}$$

where *m* is a natural number and *n* is the refractive index of the surrounding medium. The fundamental mode of the resonance corresponds to m = 1, higher dipole-like excitations correspond to odd *m*. If the diameter *D* of the antennas is no longer negligible compared to the wavelength, the resonance frequency depends also on D^{26} . Further, the skin effect has to be considered if *D* is in the range of the skin depth δ . In that case, the skin effect has an impact on the dependence of the antenna resonance on the material properties²⁷ and introduces a qualitative change of the resonance relation¹⁴. The change caused by the influence of δ on the resonance condition is opposite to the geometrical one caused by a non-negligible *D*. Through the ratio δ/D the Drude parameters influence the resonance curve of a nanoantenna. In particular, high defect density inside the metal particles would increase ω_r and thus

the ratio δ/D . In addition to the above mentioned effects, via the boundaries, the polarizability of the surrounding medium also affects the optical properties of metal nanoparticels²⁸. In order to take into account the effect of a substrate (with the refractive index ε_s) an effective medium completely embedding the wires is a good approximation. With its effective dielectric constant

$$\varepsilon_{\rm eff} = \frac{1}{2} (1 + \varepsilon_{\rm s}) \tag{4}$$

and $n = n_{\text{eff}} = \sqrt{\varepsilon_{\text{eff}}}$ the influence of the substrate on optical spectra can be described well^{22,29}.

12.2 Measurement of Antenna Resonances in the Infrared

The clearest experimental result is expected for individual gold wires with a cylindrical shape, a low defect density, and a clean surface. Gold is preferred because of its stability under ambient conditions. Gold wires with diameters between 60 and 200 nm were prepared, for example, by electrochemical deposition in etched ion-track membranes^{30,31}. Fig. 12.1 shows a scanning electron microscopy (SEM) image of a gold nanowire (with D = 100 nm), displaying its smooth contour and regular shape. Transmission electron microscopy studies proved that gold nanowires deposited under the conditions mentioned above are polycrystalline³⁰. After formation of wires with a controlled length, the polymer membrane was dissolved in dichloromethane in order to achieve clean nanowires.



Fig. 12. 1 SEM image of a gold wire with 100 nm in diameter, see text.

For IR spectroscopy these wires were transferred to different IR transparent substrates (ZnS, KBr, and CaF₂), with different refractive indices $n_{\rm s}$ ($n_{\rm ZnS}$ = 4.84 at 10 µm, $n_{\rm KBr}$ = 1.53, and $n_{\rm CaF2}$ = 1.4 at 5 µm). During the sample preparation process, the several microns long wires broke into pieces with lengths between several hundred nm and few µm, and with tip ends determined by the grain-boundary structure of the wires.

IR spectroscopic microscopy of single gold nanowires, produced as explained above, was performed at the IR beamline of the synchrotron light source ANKA (Angströmquelle Karlsruhe) at the Forschungszentrum Karlsruhe. Previous to wire-array studies, the individual wires were randomly placed on the IR transparent substrate and, therefore, their position needed to be located. At the beamline this and the length determination were done with optical microscopy with light (which gives an error of up to 8%). Then the IR-beam spot with diameter of about 8 µm was centered on the selected nanowire and IRtransmittance spectra (sample spectra) were recorded. Subsequently, in order to eliminate environmental effects like beam profile and substrate inhomogeneities, reference spectra were taken at least 10 µm away from any nanowire. The spectroscopic measurements were done with a Fourier-transform infrared (FT-IR) spectrometer (Bruker IFS 66 v/S) and a LN2-cooled mercury-cadmium telluride (MCT) detector, which collects light normally transmitted through the sample area. A small fraction of the light scattered away from the normal direction was also detected due to the collection lens (Schwarzschild objective, numerical aperture: 0.52), but its intensity was negligibly small. For both sample and reference measurements IR spectra were recorded by acquisition of at least 10 scans in the spectral range from 600 to 7000 cm-1 with a resolution of at least 16 cm-1. For further analysis of the transmitted light, a polarizer was inserted in the optical path before the sample.

For nanowires with a length of a few microns significant fundamental antenna like plasmon resonances appear in the relative transmittance spectra (ratio of the sample and reference spectra, see Fig. 12.2). Apart from the fundamental resonance, much weaker structures of resonant extinction can be found at higher wavenumbers. Usually they are attributed to higher order resonances. The fundamental resonance was observed only for electrical field parallel to the long axis of the wire (parallel polarization). For polarization perpendicular to the long wire axis (perpendicular polarization), the IR signal was below the noise level. That behaviour proves the dipole character of the excitation attributed to m = 1.



Fig. 12.2 IR transmittance spectra of two individual gold nanowires on different substrates (as indicated) normalized to the transmittance of the bare substrate. At the lower wave numbers the CaF_2 substrate arrives at the multi-phonon edge, which lowers the photon intensity at the detector, which gives rise to noise in the spectrum. The electric field was parallel to the wire.

From the measured relative transmittance spectra the ratio of the extinction cross section σ_{ext} of a single gold nanowire to its geometric cross section σ_{geo} can be estimated. As we already reported in reference 29, we use the relation

$$\frac{\sigma_{\text{ext}}}{\sigma_{\text{geo}}} = A_0 (1 - T_{\text{rel}}) \cdot (n_{\text{s}} + 1) \cdot \frac{1}{(2LD)}, \qquad (5)$$

where the substrate screening is taken into account in analogy to the normal transmittance change by a thin film on a substrate³² in comparison to a free-standing film³³. Inserting A_0 as the spot size, $(1-T_{rel})$ the transmittance change in the spectrum, n_s for the relevant substrate, L from dark-field microscopy in the visible spectral range, and D (known from wire-growth process) lead to $\sigma_{ext} / \sigma_{geo}$ as shown for one example in Fig. 12.3. Any value $\sigma_{ext} / \sigma_{geo} > 1$ means an extinction of intensity above pure shadowing, which indicates a local field enhancement in the vicinity of the nanowire. From the cross section ratio $\sigma_{ext} / \sigma_{geo}$ the spatially averaged field enhancement in the far-field ($\sqrt{\sigma_{ext} / \sigma_{geo}}$) can be calculated.



Fig. 12.3 Extinction cross section (normalized to geometric cross section): experimental curve for a gold nanowire on KBr compared to calculations for different effective refractive indices around the nanowire, see text.

The experimental spectrum that is shown in Fig. 12.3 represents the best of our measurements and reveals far-field enhancement by a factor of about 5 at the resonance maximum. The other curves were derived with two different theoretical methods. In both the methods the gold nanowire is modelled as a rod with hemispherical tip ends completely embedded in a medium (background). The far-field observation point in both calculations is in the plane defined by the rod axis and the light propagation axis. Those resonance curves marked by "FDTD calc." were modelled with the software CST Microwave studio³⁴ for antenna problems and were based on a finite difference time domain (FDTD) algorithm³⁵. In this computational approach, Maxwell's time dependent differential equations are solved over a three-dimensional grid. In our calculations one single gold nanowire was illuminated by a plane wave polarized along the long wire axis, and the antenna radiated into free space. The curves marked with "BEM calculations" were calculated with the exact boundary element method (BEM) including retardation^{14,36}. Maxwell's equations for inhomogeneous media with sharp boundaries were solved in terms of charges and currents distributed on the surfaces and interfaces. Boundary conditions are imposed via surface integrals along the boundaries between different media. Each region is characterized by a local dielectric function. The external fields interact self-consistently with the induced boundary charges and currents, which are determined by discretizing the surface integrals and solving the

appropriate matrix equations. In this approach, the scattered field due to an incident external field is calculated directly.

In the FDTD calculations, we described the metal of the gold nanowires as a lossy one. The "lossy metal" model neglects the imaginary part of the conductivity. Changing the background material from air to a dielectric one (KBr) a shift of the resonance maximum towards smaller wavenumbers is observed (dotted line). Obviously, neither the calculation based on vacuum as background nor the curve with background KBr describes the measured data. The resonance curves calculated with the boundary element method (BEM) are in better agreement with the experiment. Using an effective refractive index $n_{\text{eff}} = 1.29$ the best agreement, in strength and shape, of the resonance curve with the experimental data is obtained. Only the position of the extinction maximum differs slightly, which can be related to the error in the real length of the wire.



Fig. 12.4 Resonance wavelength versus wire length L for gold nanowires with comparable diameter on KBr. The figure shows experimental data (open symbols), idealantenna model calculations for wires embedded in different surroundings (lines), and detailed BEM calculations. The lines corresponding to Eq. 3 (m=1) are shown for comparison.

Looking at Fig. 12.4, where we display the resonance wavelengths, from the measured spectra for nanowires with comparable diameters on KBr, systematic deviation from simple antenna theory (black line) is obvious. When using the value of the refractive index of KBr in Eq .3 the calculated curve (dotted line) seems to fit our measured data. But this in conflict with the fact that the wires are not embedded in this material. On

the other hand, BEM calculations for wires embedded in an effective medium with refractive index $n_{\text{eff}} = \sqrt{\varepsilon_{\text{eff}}}$ yield good agreement with the measured data. Also the effect of a cover medium is well described by the respective $\varepsilon_{\text{eff}} = (\varepsilon_{\text{m}} + \varepsilon_{\text{s}})/2$ (for surrounding medium with dielectric constant ε_{m}) as it is shown by the coverage of individual gold nanowires on different substrates (KBr, ZnS) with paraffin wax. Due to the paraffin coverage the resonance is shifted to lower frequencies²⁹. The theoretical ratio is $n_{\text{eff}}^{\text{air}} / n_{\text{eff}}^{\text{paraffin}} = 0.875$ on KBr and 0.923 on ZnS (with $\varepsilon_{\text{m}} = 2.02$ for paraffin), respectively. These ratios are in reasonable accord with the results obtained from the spectral shift of the resonance

12.3 Surface Enhanced Infrared Absorption (SEIRA)

frequencies before and after evaporating paraffin.

Field enhancement in the IR can be exploited for surface enhanced infrared absorption (SEIRA)^{37,38} of molecules. In case of sufficiently strong near-field enhancement, SEIRA would allow the detection of vibrational fingerprints of single molecules on nanoantennas, for example, by studying light scattering with aperture-less scanning near-field microscopy³⁹.

Until now SEIRA is less well-known than SERS since its signal enhancement was measured to be only three orders of magnitudes^{37,38}. This is much less than the SERS enhancement which is many more orders of magnitude. This difference is related to the different influence of local electromagnetic field enhancement. In SEIRA the squared nearfield enhancement is relevant, whereas in SERS, which is a scattering process, two kinds of squared field enhancements are multiplied (one for the incident field and the other one for the scattered field).⁴⁰ Usually, the electromagnetic field enhancement is obtained with metal nanoparticles or nanostructured metal films. Their plasmonic resonances give rise to near-field enhancement as the main effect in SEIRA and SERS⁴¹. Other contributions to the observed enhancement of vibration signals are less important (concerning the contribution to enhancement) and less well understood. They are related to chemical interactions and non-adiabatic effects and are relevant for the first adsorbate layer only.⁴²

The near-field enhancement is strongest in the proximity of sharp edges or tips. For SERS this tip effect was already studied by several groups.^{12,43} For nanoantennas in the IR such tip effect should be much more like a lightning rod effect. This would means huge near-field enhancement at the tip ends of an antenna at resonance, much larger than

three orders of magnitude as measured as the spatially averaged enhancement for adsorbates on metal-island films compared to the same adsorbate layer without metal islands.

Acknowledgement

F. N. and A. P. gratefully acknowledge technical support by M. Süpfle, helpful discussions with A. Otto, B. Gasharova, and Y.-L. Mathis. J.A. acknowledges software access from F.J. García de Abajo, discussions with G.W. Bryant, and financial support from Gipuzkoa Foru Aldundia (Gipuzkoa Fellows program) and the Government of the Basque Country (NANOTRON project).

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